

**VIETNAM NATIONAL UNIVERSITY**  
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**ENHANCEMENTS OF CRITICAL CURRENT DENSITY  
OF  $\text{MgB}_2$  BULK AND FILM SAMPLES  
USING ION IRRADIATION TECHNIQUE**

Major: Thermal Physics

PhD Student code: 9440130.07

**ABSTRACT OF Ph.D. DISSERTATION IN PHYSICS**

**Hanoi – 2025**

The project was completed at: VNU University of Science

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The thesis can be found at:

- National Library of Vietnam;
- Library and Digital Knowledge Center, Vietnam National University, Hanoi

## ABSTRACT

Magnesium boron dioxide ( $\text{MgB}_2$ ) superconductor, discovered in 2001, offers benefits such as a high critical temperature, high current density, simple crystal structure, low cost, and additional properties like weak-link free behavior, greater coherence length, and reduced electronic anisotropy. However, its practical applications are limited due to its rapid decline in magnetic field. This dissertation focuses on improving  $J_c$  of  $\text{MgB}_2$  bulk and film samples, addressing weak link behavior in high temperature superconducting materials, requiring complex techniques.

First of all, the study aimed to fabricate  $\text{MgB}_2$  bulk samples with the strongest superconductivity performance. *Ex-situ*  $\text{MgB}_2$  powders were mixed with Mg and B, sintered at different times and temperatures. The optimal condition for fabricating  $\text{MgB}_2$  bulk samples with the highest possible  $J_c$  value was found to Pure *ex-situ*  $\text{MgB}_2 + 0.5 \text{ Mg}$  sintered at  $1000^\circ\text{C}$  for 1 hour.

Secondly, the study investigated the enhancement of  $J_c$  in  $\text{MgB}_2$  bulk sample by adding impurities, with the greatest enhancement achieved with co-additions of  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$ . The greatest  $J_c$  enhancement was obtained for the sample with the co-additions of 5 wt.%  $\text{B}_4\text{C} + 2.0 \text{ wt.}\% \text{ Dy}_2\text{O}_3$ .

Lastly, the study investigated the deposition of high-quality  $\text{MgB}_2$  thin films using hybrid physical-chemical vapor deposition (HPCVD). It was found that optimal irradiation of non-magnetic ions into  $\text{MgB}_2$  films with a dose of  $5 \times 10^{13} \text{ ion/cm}^2$  was shown to be optimal for providing the highest  $J_c$  enhancement.

## **Chapter 1: INTRODUCTION**

### **1.1. Superconductivity**

#### ***1.1.1. History and crucial milestones***

Superconductivity, a phenomenon where materials lose resistance below a critical temperature, was first observed in 1911 by Heike Kamerlingh Onnes. It led to the discovery of superconductors in various metals by the 1930s, with significant turning points including the Meissner effect, London equations, Ginzburg-Landau theory, and Josephson effect.

#### ***1.1.2. Classification***

Superconductors are categorized into type-I and type-II groups based on their behavior in a magnetic field. Type-I superconductors, composed of about thirty pure metals, exhibit the full Meissner effect, allowing field penetration even at low fields. Type-II superconductors react differently, with coherence length determining vortex radius.

### **1.2. Vortex dynamics in superconductors**

#### ***1.2.1. Flux pinning***

The Lorentz force is caused by flux vortices in type-II superconductors, causing power dissipation. To prevent dissipation, a force opposing the Lorentz force is needed. Defects in superconductors provide vortex pinning sites, and the pinning force density ( $F_p$ ) represents the total force applied to flux lines. Improving the density of defects can improve  $J_c$ .

#### ***1.2.2. Bean's critical state model***

Bean's critical state model is utilized to calculate the  $J_c$  of superconductors from magnetization hysteresis loops. Inner

filaments are protected by outer filament currents, which flow until the flux reaches the sample's core. The relationship between magnetization and  $J_c$  is represented by  $J_c = a\Delta M/d$ .

### ***1.2.3. Dew-Hughes model***

Dew-Hughes has created a model for describing flux-pinning in type-II superconductors, based on pinning center geometry and flux line interaction. The model calculates the pinning force per unit volume,  $\Delta W$  and  $\eta$ , and the efficiency factor  $\eta$ . The flux lattice stiffness determines pinning force strength. The pinning process is defined by a curve plotting normalized volume pinning force against reduced magnetic field.

## **1.3. Collective pinning theory**

Collective pinning theory is a study on the interaction between pierced vortices in type-II superconductors, focusing on field dependence of  $J_c$  and flux pinning mechanism in HTS samples. It was proposed in the 1980s, explaining how vortices form collective systems.

### ***1.3.1. Collective pinning theory for bulk superconductors***

The magnetic field induction in single vortex pinning is field independent when  $B$  is less than the crossover field, while small bundle pinning occurs when  $B > B_{sb}$ . Collective pinning theory studies type-II superconductors, vortex behavior, and thermal fluctuations.

### ***1.3.2. Collective pinning theory for thin film superconductors***

Larkin and Ovchinnikov introduced collective pinning in 1979, a method where multiple defects can significantly alter vortex dynamics. This method assumes that individual pinning forces

on the vortex line add up arbitrarily, with variations in defects' density and force determining the total force for a vortex segment. The strength of samples can be assessed using the exponent  $\beta$ .

#### **1.4. MgB<sub>2</sub> Superconductor and Subject to be discuss**

##### ***1.4.1. Basic parameters***

In 2001, Akimitsu's research group discovered superconductivity in magnesium diboride (MgB<sub>2</sub>) with a 39 K  $T_c$ , a breakthrough in superconductivity history. MgB<sub>2</sub> has a hexagonal AlB<sub>2</sub> type crystal structure with a p6/mmm space group, containing boron and magnesium layers, and a graphite honeycomb network.

##### ***1.4.2. Superconducting mechanism in MgB<sub>2</sub>***

MgB<sub>2</sub> is a phonon-mediated BCS type superconductor with a layered structure with two superconducting energy gaps. Its electronic state at the Fermi level is characterized by boron atom bonding through covalent connections and metallic linkages. The two energy gaps are caused by electron coupling in 2D bands and weak coupling in 3D.

##### ***1.4.3. Properties of MgB<sub>2</sub> Superconducting Materials***

The study explores the characteristics of MgB<sub>2</sub> superconductors, focusing on their high coherence lengths and low-temperature penetration depth, but also examining their structure due to oxygen, carbon, hydrogen, and uneven boron distribution, and their preparation methods including carbon, carbon-containing compounds, silicon carbide, titanium, tantalum, and zirconium.

##### ***1.4.4. Raman spectroscopy in MgB<sub>2</sub> superconductors***

MgB<sub>2</sub>'s superconductivity is due to electron-phonon interactions, with the E<sub>2g</sub> phonon mode being a single active mode. This Raman active mode, related to band electrical transport, can be studied using Raman scattering, providing data on crystal symmetry, particle size, and contaminants.

#### ***1.4.5. Potential application of MgB<sub>2</sub> superconductors***

High temperature superconductor (HTS) and MgB<sub>2</sub> superconductors are popular for shields, DC magnetic field generation, and small/medium power motors. They trap magnetic fields, are lightweight, and can be fabricated using techniques like hot pressing and spark plasma sintering.

Figures 1.11–1.13 show equipment for bulk MgB<sub>2</sub> production using high-pressing, hot-pressing, and spark plasma sintering techniques, producing large, stable blocks suitable for practical applications.

##### ***1.4.5.1. Trapped Magnetic Field (Quasi-Permanent Magnets)***

Magnetized MgB<sub>2</sub> and HTS bulks offer quasi-permanent magnets with high magnetic fields, making them suitable for flywheel energy storage systems. MT-YBCO bulks trap 17.24 T magnetic fields but can be destroyed by Lorentz force. MgB<sub>2</sub> bulks are simpler, less expensive, and time-efficient, suitable for various applications.

##### ***1.4.5.2. Fault Current Limiters***

Fast-operating nonlinear fault current limiters (FCLs) can limit high fault currents in power systems by increasing impedance. Superconducting materials have perfect conductivity and a quick phase transition, making them ideal for power systems. Inductive

SFCLs can protect high-voltage direct-current systems, and their circuit current and voltage drop remain unaffected by synthesis settings and ring sizes.

#### *1.4.5.3. Electrical Machines*

Superconductors replace metal wires in electrical machines, with advancements in electromagnetic characteristics enabling bulk-superconducting rotor elements.  $\text{MgB}_2$  bulk superconductor-based motors are less expensive and more efficient, with potential applications in liquid hydrogen systems.

#### *1.4.5.4. Magnetic Field Shields*

Bulk  $\text{MgB}_2$  superconductors exhibit excellent magnetic shielding properties, making them useful for passively protecting devices and orbital stations from cosmic radiation. Their basic ingredients are widely accessible and free of noble, poisonous, or rare earth elements. The shielding factors, which are based on the Hall probe position, reach their highest value around 105 at the cup's bottom.

#### ***1.4.6. Enhancements of critical current density of $\text{MgB}_2$ bulk superconductors***

$\text{MgB}_2$  is a promising material for commercial and industrial applications due to its high coherence lengths, weak-link-free behavior, and simple hexagonal crystal structure. Researchers have developed methods to produce factitious pinning centers for high magnetic field applications. *Ex-situ*  $\text{MgB}_2$  bulks were sintered at different temperatures and times, with additives like Mg and B increasing critical current density. Recent studies show

self-field  $J_c$  of  $\text{MgB}_2$  bulk samples in the range of  $2 - 4 \times 10^4$  A/cm<sup>2</sup>.

#### ***1.4.7. Enhancements of critical current density of $\text{MgB}_2$ thin-film superconductors***

##### ***1.4.7.1. Enhancements of critical current density of $\text{MgB}_2$ thin-film superconductors by using buffer layers***

The study investigates flux pinning of  $\text{MgB}_2$  films on buffered hastelloy tapes with different SiC layer thicknesses. Results show improved field dependence of  $J_c$  with increasing SiC layer thickness. Lattice mismatch between  $\text{MgB}_2$  film, ZnO buffer layer, and Hastelloy substrate significantly influences phonon behavior.

##### ***1.4.7.2. Enhancements of critical current density of $\text{MgB}_2$ thin-film superconductors by using ion irradiation***

$\text{MgB}_2$  films offer advantages like high  $T_c$ , large coherence length, simple crystal structure, low cost, and high  $J_c(0)$ . However, their practical use is limited by the quick  $J_c(B)$  fall under a magnetic field. Ion irradiation has been studied to create artificial pinning sites in superconducting materials, improving in-field  $J_c$  and  $B_{c2}$  in  $\text{MgB}_2$ , iron-based superconductors, and high- $T_c$  cuprates.

#### ***1.4.8. Subject to be discuss***

$\text{MgB}_2$  superconductor offers low cost, two energy gaps, simple crystal structure, and high self-field  $J_c$ , but its practical applications are limited due to quick  $J_c$  decline in magnetic field. The thesis aims to enhance  $J_c$  of  $\text{MgB}_2$  bulks and thin films. The overall structure of this dissertation is as follows:

- Fabrication conditions for pure  $\text{MgB}_2$  bulk superconductors showing high  $T_c$  and  $J_c$ . Adding 0.5 mol Mg and sintering at 1000 °C for 1 hour, sample S5 showed the lowest value of  $T_c$ , whereas its  $J_c$  enhancement achieved the highest value.
- Enhancing  $J_c$  of  $\text{MgB}_2$  bulk by co-addition of  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$ . The coaddition of  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$  likely improved the grain connectivity, which enhance  $B_{c2}$  and  $J_c$
- Highest  $J_c$  enhancement of  $\text{MgB}_2$ -thin films by ion irradiation. Sn ion irradiation showed an improvement in  $J_c$  at 5 and 20 K; the maximum  $J_c$  was reached for the  $5 \times 10^{13}$  atoms/cm<sup>2</sup> sample.

## Chapter 2: EXPERIMENTAL METHODS

### 2.1. $\text{MgB}_2$ bulk sample fabrications

#### 2.1.1. *Pure $\text{MgB}_2$ bulk sample fabrications*

$\text{MgB}_2$  bulk samples are prepared using *in-situ* or *ex-situ* methods. *In-situ* reactions involve melting Mg grains and diffusing them into B grains, resulting in high  $J_c$  and strong intergrain coupling. *Ex-situ*  $\text{MgB}_2$  has a higher packing factor but weaker intergrain coupling. Heat treatment improves grain-to-grain current channel and maximizes  $\text{MgB}_2$  phase production.

The study involved mixing *ex-situ*  $\text{MgB}_2$  with 0.5 mol of Mg to create Mg-added samples. In bulk  $\text{MgB}_2$  samples, 1.5 mol of Mg and 2 mol of B were added. Sintering was performed at 600 and 700 °C for 1 hour, 3 hours at 700 °C, and 1 hour at 1000 °C.

#### 2.1.2. *$\text{MgB}_2$ samples co-added $\text{B}_4\text{C}$ and $\text{Dy}_2\text{O}_3$*

Magnesium diboride ( $\text{MgB}_2$ ) powder was obtained from Alfa Aesar and mixed with 0.5 mol of Mg. Different weight

percentages of B<sub>4</sub>C were added to the MgB<sub>2</sub> samples, resulting in different samples. The mixture was ground, compressed into circular pellets, and heat-treated at 900 °C for 1 hour. The powder handling process was conducted in air. The samples are listed in Table 2.2. The process involved grinding, compression, and heat-treatment.

## **2.2. MgB<sub>2</sub> thin film sample fabrications**

### ***2.2.1. Hybrid Physical-Chemical Vapor deposition (HPCVD) system***

The hybrid physical-chemical vapor deposition (HPCVD) method was used to create MgB<sub>2</sub> thin films, maintaining thermodynamic stability at high temperatures. The process includes a load-lock chamber, heater, reactor, gas inlet, flow, pressure, temperature, and exhaust systems, using diborane as boron precursor gas and magnesium pieces as sources.

### ***2.2.2. Growth of MgB<sub>2</sub> thin films on Al<sub>2</sub>O<sub>3</sub> substrates***

The MgB<sub>2</sub> thin-film growing process involves purging a chamber with Ar gas, filling it with a susceptor, substrate, and magnesium chips. Under ambient hydrogen gas, the substrate and chips are heated inductively, forming thin coatings. Single-crystal Al<sub>2</sub>O<sub>3</sub> is used due to its hexagonal structure. The Volmer-Weber growth mode forms columnar structures, requiring superconducting samples for disorder impacts investigation.

### ***2.2.3. Ion irradiation of MgB<sub>2</sub> thin film***

A 412-nm-thick MgB<sub>2</sub> was created on c-cut Al<sub>2</sub>O<sub>3</sub> substrates using a hybrid physical-chemical vapor deposition technique. High-purity magnesium particles were heated to 700 °C, and

B<sub>2</sub>H<sub>6</sub> gas was supplied. The HUS-5SDH-2 tandem pelletron accelerator system was used for ion irradiation. Low-energy irradiation enhanced in-field  $J_c$  and  $\mu_0 H_{c2}$ , while rapid heavy-ion irradiation enhanced  $J_c$  in iron-based superconductors and high- $T_c$  cuprates.

### **2.3. Measurement methods**

#### **2.3.1. X-ray diffraction (XRD)**

X-ray diffraction (XRD) is a method used to study material crystal structures. The Bruker D8 Advance model was used to determine MgB<sub>2</sub> bulk and thin films. The study used Bragg's Law and XRD data for background determination, profile fitting, refinement, phase identification, and calculation of lattice constants.

#### **2.3.2. Physical property measurement system (PPMS)**

##### *2.3.2.1. DC susceptibility*

Susceptibility measurements were conducted using a direct current field on a surface using the Quantum Design PPMS EverCool II systems' vibrating sample magnetometer. The critical temperature ( $T_c$ ) was measured between 10 and 50 K using zerofield cooled conditions.

##### *2.3.2.2. Resistivity vs. temperature measurement*

The transport characteristics of MgB<sub>2</sub> bulk and thin films were measured using a physical property measurement system (PPMS) with a magnetic field of up to  $\pm 9$  Tesla and temperature change between 1.9 K and 400 K. The system's resistivity function was used to measure electro-transport.

##### *2.3.2.3. Magnetization vs. field measurement*

The Physical Property Measurement System (PPMS) was utilized for magnetization of  $\text{MgB}_2$  bulk and thin films, with a resolution of less than  $10^{-8}$  emu. The system, cryogen-free, used a cryostat, superconducting magnet, and measurement probes. Bean's critical state model was used to compute  $J_c$  from the magnetization hysteresis loop.

#### **2.3.3. Raman spectroscopy**

The WITec Micro-Raman Spectrometer System was used to measure Raman spectra of  $\text{MgB}_2$ 's superconducting capabilities, revealing that the  $E_{2g}$  mode's active frequency affects  $T_c$  fluctuation, confirming the BCS theory.

#### **2.3.4. Magnetic force measurement**

The study aims to determine the absolute value of the pinning force for single Abrikosov vortices and explore the spatially resolved absolute value of the London penetration depth using a special He-4 MFM with vector magnet capabilities.

### **Chapter 3: FABRICATIONS OF PURE $\text{MgB}_2$ BULK SUPERCONDUCTORS**

This chapter investigates the superconducting properties of  $\text{MgB}_2$  bulk samples under various conditions. *Ex-situ* samples were mixed with Mg and B, sintered at different times and temperatures, and Raman spectroscopy was used to investigate electron-phonon coupling (EPC). The lowest EPC value was found to be 1.094 for the  $\text{MgB}_2$  sample added with 0.5 mol Mg and sintered at  $1000^\circ\text{C}$  for one hour. The collective pinning

theory revealed that  $\delta T_c$  pinning is the predominant flux pinning mechanism across all  $\text{MgB}_2$  bulk samples.

This study investigates the effects of adding Mg and B to *ex-situ*  $\text{MgB}_2$  bulks, which were sintered at different temperatures and times, on the temperature-induced displacement ( $T_c$ ), joint conduction ( $J_c$ ), and volume pinning force ( $F_p$ ). Raman spectroscopy was used to analyze the effect of individual phonon properties on  $T_c$ . Results show that the displacement induced by Mg and B, along with sintering temperature, significantly influences  $T_c$ ,  $J_c$ , and  $F_p$  mechanisms.

### **3.1. Effect of sample fabrication conditions on $T_c$ of $\text{MgB}_2$ bulk superconductors**

The superconductivity of samples is determined by  $T_c$ , which stabilizes at 38.5 K and drops to 38.1 K for sample S5. Additions like Mg and increasing sintering temperature can reduce transition temperature, but S5 samples have the lowest  $T_c$ .

The modification of strain-induced local structural deformity and disorder significantly impacted the  $T_c$  variation of  $\text{MgB}_2$ , possibly due to incomplete reaction between Mg and B during heat treatment. *Ex-situ*  $\text{MgB}_2$  particles may decelerate phase formation. Raman spectra were used to understand  $T_c$  change in  $\text{MgB}_2$  samples.

### **3.2. Explaining the possible reason for the changes in values of $T_c$ of $\text{MgB}_2$ bulk superconductors**

$\text{MgB}_2$  superconductors'  $T_c$  variations are caused by lattice distortions and electron-phonon interactions. Raman spectrum measurements show an increase in  $E_{2g}$  peak intensity and a

decrease in half-width due to the addition of Mg and B. Lattice distortions, intense coupling, and phonon-phonon interaction may account for this shift. The addition of Mg and B reduces  $T_c$  and EPC value, leading to elevated impurity rates and significantly affecting the Raman spectrum.

### **3.3. Effect of sample preparation conditions on $J_c$ of $\text{MgB}_2$ bulk superconductors**

The study reveals that the addition of Mg and B to high-temperature superconductors enhances the flux pinning mechanism, resulting in increased  $J_c$  under various fabrication conditions. The results show that  $J_c$  is enhanced in samples S3, S4, S5, S6, and S7 at all temperatures and investigated fields. Sample S5 showed the highest improvement, while  $J_c$  descended slowly under the applied field. The study also found that adding Mg and B broadened the regime of small and large bundles, improving  $J_c$  at high external magnetic fields.

The  $J_c$  value of  $\text{MgB}_2$  bulk samples increased significantly when sintered at  $1000^\circ\text{C}$  for one hour. The  $J_c$  value reached  $1.2 \times 10^5 \text{ A/cm}^2$ , 5 times higher than previous research samples. The  $J_c$  value was 3.2 times higher at 0.5 T, 2.3 times higher at 1 T, 1.3 times higher at 1.5 T, and 1.2 times higher at 2 T.

### **3.4. Effect of sample preparation conditions on flux pinning mechanism of $\text{MgB}_2$ bulk superconductors**

The study explores the flux pinning mechanism in  $\text{MgB}_2$  bulk superconductor by adding Mg and B, focusing on improvements in  $J_c$ . The core interaction is the most efficient vortex-pinning center interaction in type-II superconductors. Two primary

pinning mechanisms exist:  $\delta l$  pinning and  $\delta T_c$  pinning. The results confirmed the supremacy of the  $\delta T_c$  pinning mechanism in all  $\text{MgB}_2$  bulks at different temperatures.

### **3.5. Effect of sample preparation conditions on geometry of pinning centers in $\text{MgB}_2$ bulk superconductors**

The study explores the impact of adding Mg and B to  $\text{MgB}_2$  bulk samples on flux pinning, focusing on the magnetic field dependence of  $F_p$ . The Dew-Hughes model was used to study supplementary flux pinning centers. The study found that the addition of Mg and B, along with temperature and sintering time, enhanced grain coupling and solid-state self-sintering, strengthening core surface pinning in  $\text{MgB}_2$  bulk superconductors.

### **Conclusion of Chapter 3**

The study investigates the local structure and superconducting properties of *ex-situ*  $\text{MgB}_2$  bulk samples under different conditions. The electron-phonon (e-ph) and superconducting characteristics of  $\text{MgB}_2$  bulk samples, such as  $T_c$ ,  $J_c$ , and  $F_p$ , were examined. The results show that  $T_c$  remains constant for samples S1–S3, declines in samples S5, and increases in samples S6–S7. The EPC constant, determined by Raman spectroscopy, significantly impacts superconducting transition behavior.  $J_c$  increasing the highest in sample S5, Pure *ex-situ*  $\text{MgB}_2 + 0.5 \text{ Mg}$  (1000 °C – 1 hour). The addition of Mg and B increased  $J_c(\text{B})$  and sintering conditions, revealing growth of small and big bundle modes. The Dew–Hughes model was used to understand the role of Mg and B addition in the flux pinning process.

## **Chapter 4: ENHANCEMENTS OF CRITICAL CURRENT DENSITY OF MgB<sub>2</sub> BULK WITH B<sub>4</sub>C AND Dy<sub>2</sub>O<sub>3</sub> ADDITIONS**

The study focuses on improving the grain coupling ( $J_c$ ) of MgB<sub>2</sub> ceramic by adding rare earth oxides (REO) and carbon (C) to create efficient flux pinning centers. Boron carbide (B<sub>4</sub>C) can be used as a precursor to form C-doped MgB<sub>2</sub> with high field-dependent  $J_c$ . ReO materials like Dy<sub>2</sub>O<sub>3</sub> can also increase  $J_c$ . The study plans to use both B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> to increase  $J_c$  of *ex-situ* MgB<sub>2</sub>.

This chapter synthesizes MgB<sub>2</sub> bulk superconductors with B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub>, confirming MgB<sub>2</sub> as the predominant phase. The properties are characterized by resistivity and magnetization measurements, with optimal additions enhancing  $B_{c2}$  and  $J_c$ .  $\delta l$  pinning is identified as the dominant mechanism.

This study investigated the effects of adding Mg, Dy<sub>2</sub>O<sub>3</sub>, and B<sub>4</sub>C to MgB<sub>2</sub> ceramics, focusing on the crystal structure and superconducting properties. The pinning mechanism was analyzed using collective pinning theory, with core interactions driving random spatial variations in transition temperature ( $\delta T_c$  pinning) and fluctuations in the charge-carrier mean free path ( $\delta l$  pinning) in high-temperature superconductors. The dominant pinning mechanism was found to be coexistence of  $\delta l$  and  $\delta T_c$  pinning. The addition of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> likely improved grain connectivity and enhanced  $B_{c2}$  and  $J_c$ .

#### **4.1. Crystal structure of MgB<sub>2</sub> bulk superconductors with co-additions of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub>**

The XRD patterns of MgB<sub>2</sub> samples with 0.5 mol of added Mg and different concentrations of Dy<sub>2</sub>O<sub>3</sub> and B<sub>4</sub>C reveal hexagonal crystal structures, possibly due to oxygen incorporation during sample preparation. B<sub>4</sub>C decreases MgB<sub>2</sub> and MgO peaks, while B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> increase them.

#### **4.2. Effect of co-additions of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> on $T_c$ of MgB<sub>2</sub> bulk superconductors**

The crystal structure of MgB<sub>2</sub> bulk with co-additions of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> is examined, revealing variations in superconducting transition and  $T_c$  values. The critical temperature ( $T_c$ ) decreases with increasing coaddition content, and the transition width broadens with increasing coaddition content. Sample S5-1 has a  $T_c$  of 38.57 K, while sample S5-4 has a  $T_c$  of 34.70 K.

#### **4.3. Effect of co-additions of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> on $\mu_0 H_{c2}$ of MgB<sub>2</sub> bulk superconductors**

The study examined the effect of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> on the upper critical field of MgB<sub>2</sub> samples. It found that the superconducting transition width increased with magnetic field strength. The optimal dopant was 5 wt.% B<sub>4</sub>C. The temperature dependence of  $B_{c2}$  showed a positive curvature, attributed to MgB<sub>2</sub>'s two-band superconductivity.

Previous studies showed that MgB<sub>2</sub> bulk samples doped with nanosized SiC, Carbon nanotube, B<sub>4</sub>C, and Dy<sub>2</sub>O<sub>3</sub> had  $B_{c2}$  values above 15 T. The MgB<sub>2</sub> + 0.5 mol Mg + 5wt.% B<sub>4</sub>C (S5-2) sample has a higher  $B_{c2}$  value.

#### **4.4. Effect of co-additions of $B_4C$ and $Dy_2O_3$ on the improvement of $J_c$ of $MgB_2$ bulk superconductors**

Figure 4.3 displays M-H hysteresis loops of samples at different temperatures, with wider loops indicating increased  $J_c$  under different manufacturing conditions.

The field dependence of magnetization of samples was measured at 10 K using modified Bean's model. The self-field  $J_c$  of each sample was determined from magnetization hysteresis loops. The pure  $MgB_2$  sample had a self-field  $J_c$  of 19 kA/cm<sup>2</sup>, while the addition of Mg (sample S5-1) increased it. Samples S5-3, S5-4, and S5-6 showed improved  $J_c$  values under magnetic fields.

The  $J_c$  value of a sample with a 5 wt.%  $B_4C$  + 2.0 wt.%  $Dy_2O_3$  coaddition improves by 1.5 – 3.2 times when subjected to a 0.1 to 2 T field at 10 K, as shown in Table 4.2.

The collective pinning theory explains vortex behavior in high-temperature superconductors by varying the influence of the magnetic field on  $J_c$ . Sample S5-5 showed increased  $B_{sb}$  and  $B_{lb}$  values, indicating the effectiveness of additional pinning centers.

#### **4.5. Effect of co-additions of $B_4C$ and $Dy_2O_3$ on flux pinning mechanism of $MgB_2$ bulk superconductors**

Core pinning is classified into two dominant types:  $\delta l$  and  $\delta T_c$ .  $\delta l$  is attributed to variations in charge carrier mean free path near defect sites, while  $\delta T_c$  is related to fluctuations in the Ginzburg–Landau parameter. The dependence of  $J_{sv}$  on reduced temperature  $t$  differs from  $\delta l$  to  $\delta T_c$  pinning. The relationship between  $B_{sb}$  and  $J_{sv}$  is  $B_{sb} = j_{sv} B_{c2}$ . The coexistence of  $\delta l$  and  $\delta T_c$  pinning mechanisms was observed in samples, similar to impurity-doped

bulk  $\text{MgB}_2$ . The role of  $\delta T_c$  pinning is more pronounced in higher-temperature regions.

#### **4.6. Effect of co-additions of $\text{B}_4\text{C}$ and $\text{Dy}_2\text{O}_3$ on geometry of pinning centers in $\text{MgB}_2$ bulk superconductors**

The study examined the impact of adding  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$  to  $\text{MgB}_2$  bulk samples, focusing on their pinning properties. The Dew-Hughes model predicted normal core-surface and core point pinning, with core-correlated pinning being dominant. The addition of  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$  may enhance connectivity between  $\text{MgB}_2$  grains.

#### **Conclusion of Chapter 4**

The study investigates the improved flux pinning properties of  $\text{MgB}_2$  bulk samples doped with  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$ , revealing small lattice distortions. The greatest increase in  $J_c$  was observed in the sample with co-addition of 5 wt.%  $\text{B}_4\text{C}$  and 2.0 wt.%  $\text{Dy}_2\text{O}_3$ . The greatest increase in  $B_{c2}$  was observed in the sample with 5 wt.%  $\text{B}_4\text{C}$ . The dominant pinning mechanism was  $\delta I$  pinning, with two-dimensional pinning centers associated with grain boundaries. These findings can be used to enhance manufacturing parameters for  $\text{MgB}_2$  bulk samples.

### **Chapter 5: ENHANCEMENTS OF CRITICAL CURRENT DENSITY OF $\text{MgB}_2$ THIN FILM WITH $\text{Sn}^{2+}$ ION IRRADIATED**

Ion irradiation significantly impacts in-field  $J_c$  and  $B_{c2}$  of  $\text{MgB}_2$ , with high- $T_c$  cuprate superconductors often improving by columnar failures along ion tracks. Defects in type-II

superconductors disrupt vortex movement, increasing essential superconducting characteristics like in-field  $J_c$  and  $B_{c2}$ . Understanding faults' impact on superconductivity is crucial for scientific development and technological adoption.

This chapter discusses the effects of 2 MeV  $\text{Sn}^{2+}$  irradiation on  $\text{MgB}_2$  films grown on an  $\text{Al}_2\text{O}_3$  substrate. The study used a hybrid physical-chemical vapor deposition technique and the HUS-5SDH-2 tandem pelletron accelerator system to expose 2 MeV energetic Sn ions to different doses, determining their mean projected ranges and damage events.

Raman spectroscopy was used to investigate changes in  $T_c$  and  $B_{c2}$  in  $\text{MgB}_2$  films, a superconductor facilitated by phonons. The study found that ion irradiation displacement is essential for modifying  $T_c$ , in-field  $J_c$ , and  $B_c$ . The study also examined the relationship between EPC from Raman analysis and superconductivity properties, revealing the relationship between  $B_{c2}$  and thermodynamic critical fields.

### **5.1. Effect of $\text{Sn}^{2+}$ ion irradiation on $T_c$ and RRR of $\text{MgB}_2$ thin films**

The study examined the temperature dependence of samples' resistivity to determine their superconductivity. Results showed that all samples exhibited metallic behavior in high temperatures, with a superconducting transition occurring as temperature steadily decreased. The irradiated samples showed a decrease in resistivity ( $T_c$ ) due to lattice distortions in  $\text{MgB}_2$  films. Ion irradiation altered  $T_c$  by altering local texture and strain-induced

structural modification, such as deformity and clutter of the crystal lattice.

The residual-resistance ratio (RRR) measures the purity and structure of a superconductor based on temperature dependence. The 7E13 sample had the lowest  $T_c$  value at 30.47 K, with a decrease in RRR values with increasing ion dose, possibly due to lattice disorder after irradiation.

### **5.3. Effect of $\text{Sn}^{2+}$ ion irradiation on electron-phonon coupling properties of $\text{MgB}_2$ thin film superconductors**

Raman scattering is used to study the  $E_{2g}$  phonon mode in  $\text{MgB}_2$ , a phonon-mediated superconductor. The Raman spectrum of  $\text{Sn}^{2+}$  irradiated films shows peaks at  $750\text{ cm}^{-1}$  and  $792\text{ cm}^{-1}$ , with weak influence on critical temperature. As irradiation dose increases, phonon density of states peaks increase. The relationship between Raman signals and superconducting properties is explained by the McMillan formula and Bardeen–Cooper–Schrieffer theory.

### **5.4. Effect of $\text{Sn}^{2+}$ ion irradiation on the improvement of $J_c$ of $\text{MgB}_2$ thin film superconductors**

The study explores the enhancement of in-field  $J_c$  in  $\text{MgB}_2$ -thin films at 10 K. The pristine films had a substantial  $J_c$  at zero field, which disappeared quickly in magnetic fields.  $\text{Sn}^{2+}$  ion irradiation significantly improved the in-field  $J_c$ , matching the pristine sample's value at 1.33 T. Sample 5E13 showed the best  $J_c$  value at 10 K, with an irreversible field increasing more than three times from 2.23 to 6.49 T.

The  $J_c$  value of  $\text{MgB}_2$  thin film samples was found to be  $2.5 \times 10^6$  A/cm<sup>2</sup> when irradiated with  $\text{Sn}^{2+}$  ions at a dose of  $5 \times 10^{13}$  ions/cm<sup>2</sup> and 2 MeV, a 2.5 times higher value than previously reported values.

### **5.5. Effect of $\text{Sn}^{2+}$ ion irradiation on $B_{c2}$ of $\text{MgB}_2$ thin film superconductors**

The study explores the effect of  $\text{Sn}^{2+}$  irradiation on the  $B_{c2}$  of  $\text{MgB}_2$  samples in a magnetic field. It finds that Sample 5E13 is the optimal fluence for increasing  $B_{c2}$ , while pristine samples have a pristine  $B_{c2}(0)$  of 5.889 T.  $\text{Sn}^{2+}$  irradiation enhances  $B_{c2}(0)$  but suppresses  $T_c$  suppression. The study suggests that a superconducting layer enhances superconducting characteristics while reducing  $T_c$ .

Previous studies showed that  $\text{MgB}_2$  films irradiated with He and Co ions had  $B_{c2}$  values of 8.9 and 9.5 T, respectively.  $\text{Sn}^{2+}$  ion irradiation, at a dose of  $5 \times 10^{13}$  ions/cm<sup>2</sup> and 2 MeV, had a  $B_{c2}$  value of 12.941 T.

### **5.6. Observations of magnetic vortex cores $\text{MgB}_2$ thin films before and after $\text{Sn}^{2+}$ ion irradiation**

The study used a homemade  $^4\text{He}$  probe to measure magnetic force gradients in superconducting samples. Homogeneous vortices were observed in Nb and pristine  $\text{MgB}_2$  samples, while sample 5E13 showed inhomogeneity. The Meissner response force gradient was used to screen the magnetic field. The pristine and 5E13 samples had larger penetration depths.

### **5.8. Comparison of doping effect on the $J_c$ of MgB<sub>2</sub> bulk superconductor and ion irradiation effect on the $J_c$ of MgB<sub>2</sub> thin films**

The study systematically examined the impact of fabrication conditions on the  $J_c$  of MgB<sub>2</sub> bulk samples, revealing that pure ex-situ MgB<sub>2</sub> + 0.5 Mg sintered at 1000 °C for 1 hour increases  $J_c$  the most. The study also explored the effect of doping on  $J_c$ , with the greatest enhancement achieved with 5 wt.% B<sub>4</sub>C + 2.0 wt.% Dy<sub>2</sub>O<sub>3</sub>. The research also focused on MgB<sub>2</sub> thin film superconductor, with the strongest increase observed when irradiated with Sn<sup>2+</sup> ions.

### **Conclusion of Chapter 5**

The study explains variations in electron-phonon and superconductivity properties in MgB<sub>2</sub>-thin film samples by irradiation with Sn<sup>2+</sup> ions. The study found a high correlation between the E<sub>2g</sub> mode and  $T_c$ , suggesting the desired E<sub>2g</sub> mode in MgB<sub>2</sub>. The behavior of superconducting transitions is significantly influenced by the EPC constant, suggesting disorder strength influences the superconducting process of MgB<sub>2</sub>. The value of  $T_c$  gradually decreased, whereas the values of  $J_c$  and  $B_{c2}$  increased and reached a maximum at 5E13. Moreover, the opposition between the high value of  $\mu_0 H_{irr}$  and low value of  $\beta$  can improve  $J_c$  values. The study suggests MgB<sub>2</sub>-thin films irradiated with Sn<sup>2+</sup> ions are attractive for power applications.

## CONCLUSION

The present dissertation delves into the investigation of the critical current density and flux pinning mechanism in  $\text{MgB}_2$  superconductors. The main findings of this dissertation, which include advances in  $J_c$  in  $\text{MgB}_2$  superconductors, can be summed up as follows:

Initially, the fabrication of  $\text{MgB}_2$  bulk samples was optimized by using the conventional solid state reaction technique. By adjusting the additions of starting materials of Mg and B as well as the sintering temperature, critical temperature ( $T_c$ ), critical current density ( $J_c$ ), and the flux pinning force density ( $F_p$ ) of the  $\text{MgB}_2$  bulk were found to vary. The variation of  $T_c$  was found to link to the corresponding variation in electron-phonon coupling and the formation of phonon density of state of the samples probed by Raman spectra. The  $J_c$  enhancements were attributed to the improvements of grain connectivity, which was believed to effectively worked as 2D pinning centers by using the Dew-Hughes model. Therefore, the optimal sample fabrication condition to increase  $J_c$  was  $\text{MgB}_2 + 0.5 \text{ Mg}$  sintered at  $1000^\circ\text{C}$  for 1 hour.

To the further enhancements of  $J_c$  of  $\text{MgB}_2$  bulk, the additions of proper impurity into solid-state reacted  $\text{MgB}_2$  bulk were carried out. By applying the optimal fabrication conditions of  $\text{MgB}_2$ , the improved flux-pinning properties of  $\text{MgB}_2$  bulk samples with co-additions of  $\text{B}_4\text{C}$  and  $\text{Dy}_2\text{O}_3$  were investigated. The greatest  $J_c$  enhancement of  $\sim 1.5 - 3.2$  times in the range of 0.1 to 2 T at 10 K was obtained for the sample with 5 wt.%  $\text{B}_4\text{C} + 2.0 \text{ wt.}\%$

Dy<sub>2</sub>O<sub>3</sub>. More interestingly, the co-additions of B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> were shown to provide the enhancements of  $B_{c2}$ , with the maximum  $B_{c2}$  was achieve for the 5 wt.% B<sub>4</sub>C added MgB<sub>2</sub> bulk sample up to 1.6 – 1.7 times. The improvements of grain connectivity and the dominance of 2D pinning centers in B<sub>4</sub>C and Dy<sub>2</sub>O<sub>3</sub> added MgB<sub>2</sub> bulk were also revealed by using the Dew-Hughes model. The evidences were likely to be reasons for enhancements of  $J_c$  and  $B_{c2}$ .

To further satisfy the power related application of MgB<sub>2</sub>, the nearly single crystalline MgB<sub>2</sub> thin films were fabricated by using HPCVD technique. The  $J_c$  of the MgB<sub>2</sub> films was found to be much higher than that of the MgB<sub>2</sub> bulk and the dominant pinning mechanism was still identified to be 2D pinning centers. The addition of different kind of pinning centers was performed by using the ion irradiation technique. The Sn<sup>2+</sup> ions were irradiated into and expected to thoroughly pass the MgB<sub>2</sub> films. Evidence for the formation of point-like defects in form of ion track or crystal imperfections were indicated by the shifts in the XRD and Raman E<sub>2g</sub> peaks. By varying the Sn<sup>2+</sup> ion dose from  $2 \times 10^{12}$  to  $7 \times 10^{13}$  ion/cm<sup>2</sup>, the obvious enhancements of  $J_c$  and  $B_{c2}$  were reached, those were ~ 2.5 – 1000 times in the range of 0.1 to 2 T at 10 K and 1.4 – 1.5 times, respectively. The pinning strength - obtained by using the collective pinning theory - of the Sn<sup>2+</sup> irradiated MgB<sub>2</sub> films was found to be improved. The pinning mechanism was slightly shift from 2D to 1D pinning as suggested by the Dew Hughes model, which was possibly originated from the point-like defects. The researches on

irradiation process to improve critical parameters of superconductors might be expanded with different kinds of ions.

## DISSERTATION PUBLICATIONS

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